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Addition of Nanohybrids Particles and Fiber to Heat Cured PMMA Denture Base Materials

Intisar J. Ismail^{1*}, Balkees M. Diya² and Farah J. Ebrahim²

¹ College of Dentistry, Al-Bayan University, Baghdad, Iraq
 ² Department of Applied Science, University of Technology, Baghdad, Iraq
 *Corresponding e-mail: <u>intesaralrawi@yahoo.com</u>

ABSTRACT

This study investigated the effect of the addition of nanohybrid particles (nano lithium silicate, nanoparticles highdensity polyethylene and nanoparticles zirconium oxide) with fiber reinforcement (ultra- high-density polyethylene fiber) on transverse, impact strength and surface hardness of heat cured PMMA. A total of 60 specimens were prepared, 40 specimens of each group were fabricated with dimensions of $(65 \times 10 \times 2.5)$ mm to conduct the transverse strength and hardness tests, while the remaining 20 specimens of each group were fabricated with dimensions of $(80 \times 10 \times 4)$ mm to perform the impact strength test. ZrO_2 nano-fillers were surface-treated with a saline coupling agent to provide bonding between ZrO_2 nanoparticles and PMMA matrix. DC-glow discharge system was utilized for plasma surface treatment to ultra-high density polyethylene fibers (UHDPE) to change fiber surface. The results showed that transverse strength and surface hardness begin to decrease after an increase in the fillers percentage and an extreme rise in the impact strength by the increasing fillers percentage in composites.

Keywords: Nanohybrid composites, Heat cured PMMA, Fiber reinforcement

INTRODUCTION

Composites structure is affected by the aesthetic, physical, mechanical properties and their clinical behavior [1]. Chemical composition of dental composite are classified into organic matrix or organic phase, inorganic matrix (filler or disperse) and organosilane or coupling agent to bond filler with the organic resin are the. This agent is a molecule with silane groups at one end (ion bond to SiO2) and methacrylate groups at the other (covalent bond with the structure of composites affected on aesthetic, physical, mechanical properties and their clinical behavior) [1,2].

Composites like all dental materials have their own restrictions, such as cavity formalization due to polymerization contraction through the setting, which causes leakage and marginal discoloration. Though, by compression with ceramic many composites have higher wear resistance than ceramic. In addition, some of them have wear rates similar to amalgam [3].

Mechanical properties of the composite resin are affected by several factors, like light source, chemical composition, mode of photoactivation, light source and distance of emitted radiation. Compressive strength, flexural strength, and modulus strength can be affected by increasing the amount of nanocomposite inorganic filler [4].

Nanocomposites are generally of two types: nanohybrid types and nanofill types. Nanohybrids comprises of milled glass fillers along with nanoparticles in the size range of 40-50 nm. The nanofill types of composites are formed by a combination of nanomers, which are nanosized mono-dispersed, non-aggregated silica filler particles in the size range of 20-75 nm, and nanoclusters, which are agglomerations of a combination of zirconia-silica and silica nanomeric particles.

The typical "non-agglomerated" property in nanocomposites is achieved by utilizing the coating process during particle manufacturing process. This ensures even distribution of the discrete filler particles within the matrix with no clumping or sticking of particles, thus ensuring a smoother consistency and improved flow [5-6].

MATERIALS AND METHODS

Several materials used for denture base are: PMMA (Hot cure as pour type denture base materials, PMMA, MMA, Spofa Dental a. s., Czech). The mold materials are made from (di stone, pink, Bulgaria). High density polyethylene particles nano filler, (type: 10nm, particle), MPCC COM. Lithium silicate nano filler, liquid, pH=11.47, density=1.181, $SiO_2=19.6\%$, $Li_2O=2.02\%$, XINSEN YVANCO., ltd. Zirconium Oxide (ZrO₂) nano- filler, (type: 70-80nm, nanoparticle, 99.9%HW, china. Polyethylene fibers (ultra-high density weight, white in color) mat, Cemfiber[®] china. TMSPM (tri-methacryloxy propyl trimethoxy saline) silane coupling agent.

Filler Treatment and Sample Preparation

Two methods were used in filler treatment which includes:

1. Plasma Surface Treatment: The treatment of ultra-high density polyethylene fibers using DC-glow discharge system was done by using a device called a DC-glow discharge system. In this study, the time of treatment of UHDP fibers was 10 minutes. The condition of the plasma system is shown in Table 1.

Variables	Value
Gas name	0 ₂
Gas pressure	7×10^{-1} mbar
Voltage	600 Volt
Current	30 mamp
Plasma treated time	10 min
Distant between anode and cathode	4 cm

Table 1 Condition of plasma system

The polyethylene fibers to be treated were placed in the center of cathode assembly of the glow-discharge plasma system. The space between the electrodes was 4 cm and the voltage was set to produce enough plasma depending on the sample size. Plasma chamber was evacuated to pressure 7×10^{-1} mbar using a mechanical rotary pump when the plasma chamber evacuated the oxygen gas was introduced in the chamber. The fiber was sputter cleaned until all micro arcs disappear and uniform glow could be noticed through the chamber, as shown in Figure 1.



Figure 1 Producing plasma in the plasma chamber were the UHDPE fiber between the anode and cathode

2. Silanation of Nanofillers (ZrO₂): The formation of reactive groups onto the surface of the fillers was achieved by reaction of the (TMSPM) silane coupling agent with (tri- methyacryloxy propyl trimethoxy saline) with zirconium oxide fillers.

Total 30 gm of nanofiller was dispersed in 200 ml of toluene by stirring with a magnetic stirrer for 20 mins into a flask

Ismail, et al.

and then sonicated at ambient temperature. Then 1.5 gm of silane (5.0 wt.% to nano-filler) was applied dropwise by sterile syringe under quick stirrer. With parafilm, the flask was wrapped and for 48 hours the slurry was left positioned in the flask. Toluene was separated by rotary evaporator under vacuum at 60°C, at rotary 150 rpm for 30 minutes. Then nano-filler was kept at room temperature after use [7].

Specimens Grouping

In this study 60 specimens were prepared; according to tests used the specimens were divided to 4 groups, one for control and another for all additions (nano LiSiO_3 + nano ZrO_2 + nano HDPE + UHDPE fibers), where each subgroup contains 5 specimens to be tested.

Test Specimens General Preparation

Acrylic specimen preparation included preparation of plastic model, proportioning and mixing of the acrylic, addition of filler, packing, curing and finishing and polishing. The composites were prepared according to the percentages in Table 2.

Table 2 Polymer, monomer and fillers (nano ZrO₂, nano HDPE, nano Li₂SiO₃ and UHDPE fibers) percentages and amounts utilized in the study

Li,SiO,	Amount of	ZrO,	Amount of	HDPE	Amount of	UHPPEF	Amount of	Amount of	Amount of
Percentage	Li ₂ SiO ₃	Percentage	ZrO ₂	Percentage	HDPE	Percentage	UHDPEF	PMMA	monomer
0.000	0.00 gm	0.000	0.00 gm	0.000	0.00 gm	0.000	0.00 gm	22.00 gm	10 ml
0.005	0.11gm	0.005	0.11 gm	0.010	0.22 gm	0.010	0.22 gm	21.34 gm	10 ml
0.500	0.11gm	0.010	0.22 gm	0.015	0.33 gm	0.015	0.33 gm	21.01 gm	10 ml
0.500	0.11gm	0.015	0.33 gm	0.020	0.44 gm	0.020	0.44 gm	20.68 gm	10 ml

Mechanical Tests Utilized to Examine Properties

Evaluations of the mechanical properties of the prepared monohybrid composite denture base were compared with conventional denture base (heat cure acrylic resin).

Transverse strength: The specimens used were prepared as described with dimensions (65 mm \times 10 mm \times 2.5 mm \pm 0.2 mm). Five specimens of each concentration make a total of 20 specimens for measurements of transverse strength. All the specimens were immersed in distilled water at 37°C for 48 hours before being tested (ADA No. 12, 1999) [8].

The test was performed using a universal testing machine (WDW-200E), each specimen was positioned on the bending fixture which consists of two parallel supports (50 mm apart), the maximum load was 50 Kg and the load was applied across heads speed of 1 mm/min by a rod placed between the supports making deflection until fracture occurs. Transverse strength was calculated using the following equation:

 $T = \frac{3PL}{2bd^2}$ (ADA specification No. 12, 1999) Where:

T: Transverse strength (N/mm²)

P: Maximum force exerted on samples (N)

- L: Distance between the supports;
- b, d: The width and depth of samples (mm) respectively

Impact strength test: The specimens used were prepared as described with dimensions ($80 \text{ mm} \times 10 \text{ mm} \times 4 \text{ mm} \pm 0.2 \text{ mm}$), international standard (ISO 179, 2000) for unnotched specimens [9]. Five specimens of each concentration were prepared to make a total of 20 specimens for impact strength measurements. Specimens were tested after being conditioned in distilled water at 37° C for 48 hours.

Impact strength test was conducted following the procedure is given by the ISO 179 with charpy type impact testing instrument. The specimen was supported horizontally at its ends and struck by a free-swinging pendulum which was released from a fixed height in the middle. A pendulum of 2 joules testing capacity was used. The scale reading gave

the impact energy absorbed to fracture the specimen in joules when struck by a sudden blow. The charpy impact strength of unnotched specimen was calculated in KJ/mm² as given by the following equation:

Impact strength=
$$\frac{E}{b.d} \times 10^3$$

Where:

- E: The impact energy in Joules
- b, d: Width and thickness, respectively

Surface Hardness Test

Specimens of acrylic resin were prepared with a dimension (65 mm x 10 mm x 2.5 ± 0.1 mm) (ISO 179, 2000) for the unnotched specimen. Specimens were stored in distilled water at 37°C for 48 hours before being tested (ADA specification No. 12, 1999).

Surface hardness was determined using durometer hardness tester from type (shore D) that was fabricated by Time group INC company according to American National Standard/American Dental Association (ANS/ADA) No. 12, 1999 which is suitable for an acrylic resin material. The instrument consists of blunt-pointed indenter 0.8 mm in diameter that tapers to a cylinder 1.6 mm. The indenter is attached to a digital scale that is graduated from 0 to 100 units. The usual method is to press down firmly and quickly on the indenter and record the maximum reading as the shore "D" hardness, measurements were taken directly from the digital scale reading. About 3 measurements were done on different areas of each specimen (the same selected area of each specimen), and an average of 3 reading was calculated.

Statistical Analysis of Data

The data obtained were first entered in the computer, then tabulated and subjected to statistical evaluation. Using descriptive statistics for mean and standard deviation was then analyzed by ANOVA to study the differences between groups. All tests were performed at a significance level of 5% in SPSS version 20.0 software.

RESULTS

FTIR was done for zirconium oxide nanofillers before salinization to determine the absorption bands of the compound. Also, another FTIR was done to zirconium oxide nanofillers after salinization to evaluate the differences in active groups, as seen in Figures 2, 3.



Figure 2 FTIR for nanoparticles ZrO₂ before salinization



Figure 3 FTIR for nanoparticles ZrO₂ after salinization

The stretching band at 1460cm⁻¹, and for the double bond between (CH_2-CH_3) which occurred at 1396cm⁻¹ with a group of two peaks between 624-700cm⁻¹ may be caused by stretching of (C=O) bond. Comparison between the Figure 4 and 5, in Figure 5 IR spectrum of the composite after increment fillers to PMMA, showed a change in shape of absorption peaks, at 3437cm⁻¹ O-H stretch and H- bond also Si-OH. In 1732cm⁻¹ C=O stretch, 1643.35cm⁻¹ -C=C- stretch. At 1192cm⁻¹ H-wag, 1086 cm⁻¹ Si-O-Li, while 840.96-914.26cm⁻¹ SiO stretch.



Figure 4 FTIR for control PMMA



Figure 5 FTIR for composite

Mean values of transverse strength were present in Figure 6 and Table 3,4 for prepared composites specimens. The calculations revealed that the transverse strength decreased at $\text{Li}_2\text{SiO}_3(0.5\%) + \text{ZrO}_2(1.5\%) + \text{HDPE}(2\%) + \text{UHDPE}(2\%)$ for composites (74.55 MPa). But it is statistically not significant when applying ANOVA test.



Figure 6 Bar chart represent mean value for transverse strength (MPa) of the composite contain (nano Li_2SiO_3 , nano ZrO_2 , nano HDPE particles and UHDPE fiber)

Table 3 Descriptive statistics for transverse strength (MPa) of the composite contain (nano Li2SiO3, nano ZrO2, nano
HDPE particles and UHDPE fiber)

Groups	Ν	Mean	Std. Deviation	Std. Error
Control	5	60.0580	8.867672	2.03060
Li ₂ SiO ₃ (0.5%) + ZrO ₂ (0.5%) + HDPE (1%) + UHDPE (1%)	5	56.6992	5.276300	2.60387
$Li_2SiO_3 (0.5\%) + ZrO_2 (1\%) + HDPE (1.5\%) + UHDPE (1.5\%)$	5	54.6358	4.833940	0.78291
Li ₂ SiO ₃ (0.5%) + ZrO ₂ (1.5%) + HDPE (2%) + UHDPE (2%)	5	53.9302	4.631657	0.78291

Table 4 ANOVA results for transverse strength (MPa) of the composite contain (nano Li₂SiO₃, nano ZrO₂, nano HDPE particles and UHDPE fiber)

ANOVA	Sum of Squares	df	Mean Square	F-value	Sig.
Between groups	2.306	3	0.769	0.051	0.984
Within groups	242.587	16	15.162	0.051	

For impact strength test, the results of this study indicated that, improved impact strength occurred in composite specimens at concentration fillers (0.5% Li₂SiO₃ + 1.5% ZrO₂ + 2% HDPE + 2% UHDPE fibers) (11.1980 Kj/m²) investigated in Tables 5, 6 and Figure 7.

Table 5 Descriptive statistics for Impact strength (kJ/m²) of the composite contain (nano Li₂SiO₃, nano ZrO₂, nano HDPE Particles, and UHDPE fiber)

Groups	Ν	Mean	Std. Deviation	Std. Error
Control	5	5.140	0.72319	0.32342
$Li_2SiO_3(0.5\%) + ZrO_2(0.5\%) + HDPE (1\%) + UHDPE (1\%)$	5	9.416	1.44412	0.64583
$Li_2SiO_3 (0.5\%) + ZrO_2 (1\%) + HDPE (1.5\%) + UHDPE (1.5\%)$	5	10.050	1.26590	0.56613
Li ₂ SiO ₃ (0.5%) + ZrO ₂ (1.5%) + HDPE (2%) + UHDPE (2%)	5	11.198	0.69139	0.30920

Table 6 ANOVA results for Impact strength (kJ/m²) of the composite contain (nano Li₂SiO₃, nano ZrO₂, nano HDPE particles, and UHDPE fiber)

ANOVA	Sum of Squares	df	Mean Square	F-value	Sig.
Between Groups	104.984	3	34.995	20.952	0.000
Within Groups	18.756	16	1.172	29.855	



Figure 7 Bar chart represent mean value impact strength (kJ/mm²) of the composite contain (nano Li²SiO³, nano ZrO², nano HDPE particles, and UHDPE fiber)

A decrease in the surface hardness for composite specimen was prepared with $(0.5\% \text{ Li}_2\text{SiO}_3 + 1.5\% \text{ ZrO}_2 + 2\% \text{ HDPE} + 2\% \text{ UHDPE}$ fibers) of the mean (86.4860) when applying ANOVA test between groups (Table 7,8 and Figure 8).

Table 7 Descriptive statistics for Hardness number of the composite contain (nano Li₂SiO₃, nano ZrO₂, nano HDPE particles, and UHDPE fiber)

Groups	Ν	Mean	Std. Deviation	Std. Error
Control	5	81.072	1.21857	0.54496
$Li_{2}SiO_{3}(0.5\%) + ZrO_{2}(0.5\%) + HDPE(1\%) + UHDPE(1\%)$	5	87.496	2.15529	0.96388
$Li_{2}SiO_{3}(0.5\%) + ZrO_{2}(1\%) + HDPE(1.5\%) + UHDPE(1.5\%)$	5	87.218	0.67622	0.30241
$Li_{2}SiO_{3}(0.5\%) + ZrO_{2}(1.5\%) + HDPE(2\%) + UHDPE(2\%)$	5	86.486	0.85410	0.38196



 Table 8 ANOVA test results for Hardness number of the composite contain (nano Li₂SiO₃, nano ZrO₂, nano HDPE particles, and UHDPE fiber)

Figure 8 Bar Chart of Mean Surface Hardness of the composite contains (nano Li₂SiO₃, nano ZrO₂, nano HDPE particles, and UHDPE fiber)

DISCUSSION

In this study, the results showed a decrease in flexural strength by adding $(0.5\% \text{ Li}_2 \text{SiO}_3 + 1.5\% \text{ nZrO}_2 + 2\% \text{ nHDPE}$ particles + 2% nUHDPE fibers) to PMMA resin as illustrated in Figure 6. This can be attributed to the degree of nanoparticles dispersion in the PMMA resin matrix and the chemical composition of the commercial acrylic resin material used in the study. From this study, it can be concluded that the polar interactions between the nanoparticles and C=O of polymer matrix (weak interactions) are at adequate low concentrations that results in improved flexural strength. At higher concentrations, nanoparticles may also act as impurities of polymerization that causes a decrease in the degree of cure (DC). Hence, more amount of unreacted monomer may be left over within the matrix that results in decreasing the flexural strength [10]. The decrease in flexural strength may be related to the decrease in the cross-section of the load-bearing polymer matrix, stress concentration because of too many filler particles, changes in the modulus of elasticity of the resin and mode of crack propagation through the specimen due to an increased amount of fillers, void formation from entrapped air and moisture, incomplete wetting of the fillers by the resin, and the fact that nanoparticles act as an interfering factor in the integrity of the polymer matrix [11]. Furthermore, the decrease may be caused by the difficulty of obtaining good adhesion between polyethylene fibers and the resin matrix, or by random orientation of fibers in the specimens of most groups [12].

In Figure 7, an increase in the impact strength by adding hybrid composite $(1.5\% nZrO_2 \text{ particles} + 0.5\% nLi_2SiO_3 + 2\% nHDPE \text{ particles} + 2\% UHDPE fibers)$ was observed. This is due to the increase in the filler load achieved which caused a reduction in the dimensions of the particles, along with their wide size distribution. This consequently reduces the polymerization shrinkage and increases the mechanical properties [13]. The increase may be resisted by the presence of fillers. Fillers intersect microcracks and bridge the gap between two surfaces of the crack. Under loading condition, when a crack starts to propagate, the fibers apply force opposing the crack propagation due to which the strength increases. Interface force of fiber-matrix is directly proportional to the surface area between the matrix and fiber, thus fiber concentration plays a major role in enhancing the strength quality. Fiber-matrix debonding may lower the strength and stiffness to the undamaged structural fiber reinforced polymers [14].

In Figure 8, it can be noted that the hardness increases and then starts to decrease. This is probably due to grain size effects and the reinforcement role of nanoparticles [15]. A low hardness value of a resin composite indicates poor chemical/physical bonding between the matrix and filler interface [13]. The decrease in the hardness is due to the increase in the filler ratio which causes an increase in the cluster of the nano-filled composite and causes consequently a decrease in the hardness value of the composite [16]. Further then, comparatively uniform distribution of all

phases and the lowering of interparticle distance with the matrix will result in raising the resistance of composites to indentation [17].

CONCLUSION

The addition of four different fillers to heat cured PMMA increases the impact strength, while, the transverse strength and surface hardness decreases by an increase in the percentage of fillers. There is a difference in the peak of FTIR between the control PMMA and the composites when fillers are added to PMMA.

DECLARATIONS

Conflict of Interest

The authors declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article.

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